



Materials Testing 2.0 for Creep: Accelerating High Temperature Materials Qualification

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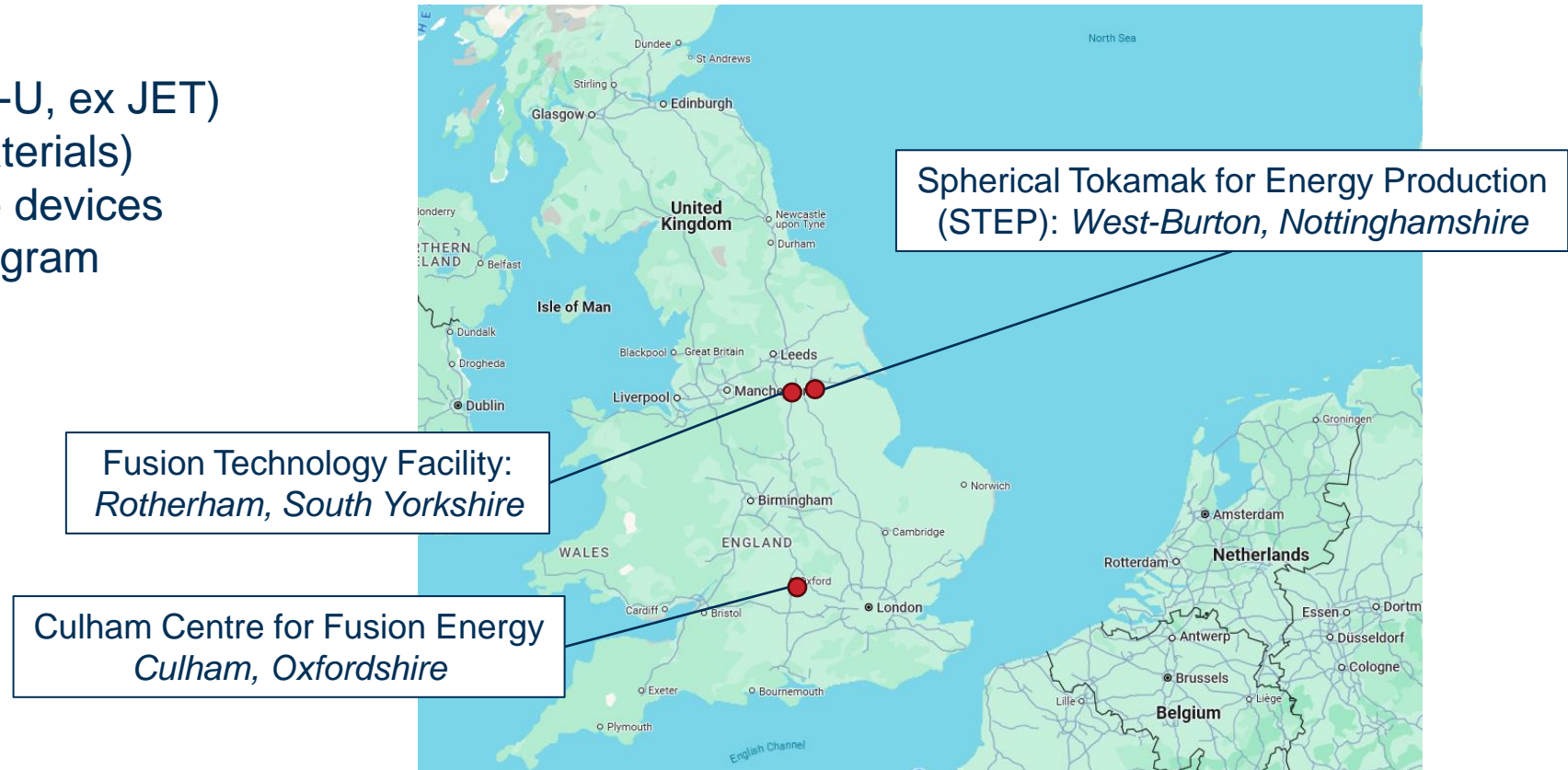
UK Atomic Energy Authority (UKAEA)

Mission:

“To lead the delivery of sustainable fusion energy and maximise the scientific and economic benefit”

Activities:

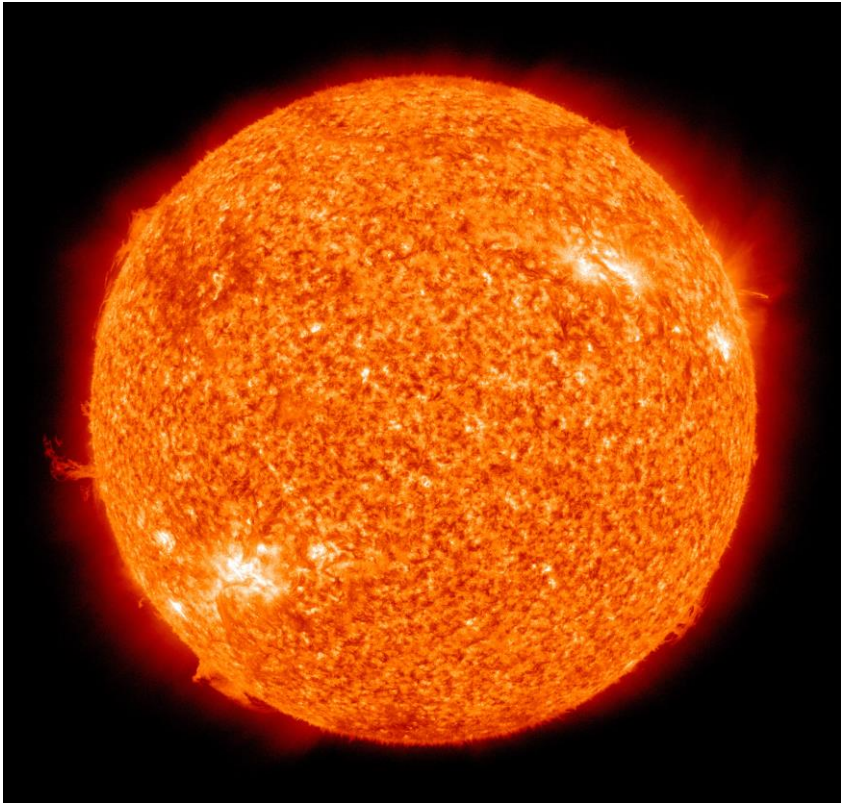
- Operate fusion devices (MAST-U, ex JET)
- Research (plasma physics, materials)
- Develop components for future devices
- Contribute to EUROfusion program
- Robotics
- High performance computing



Fusion Energy

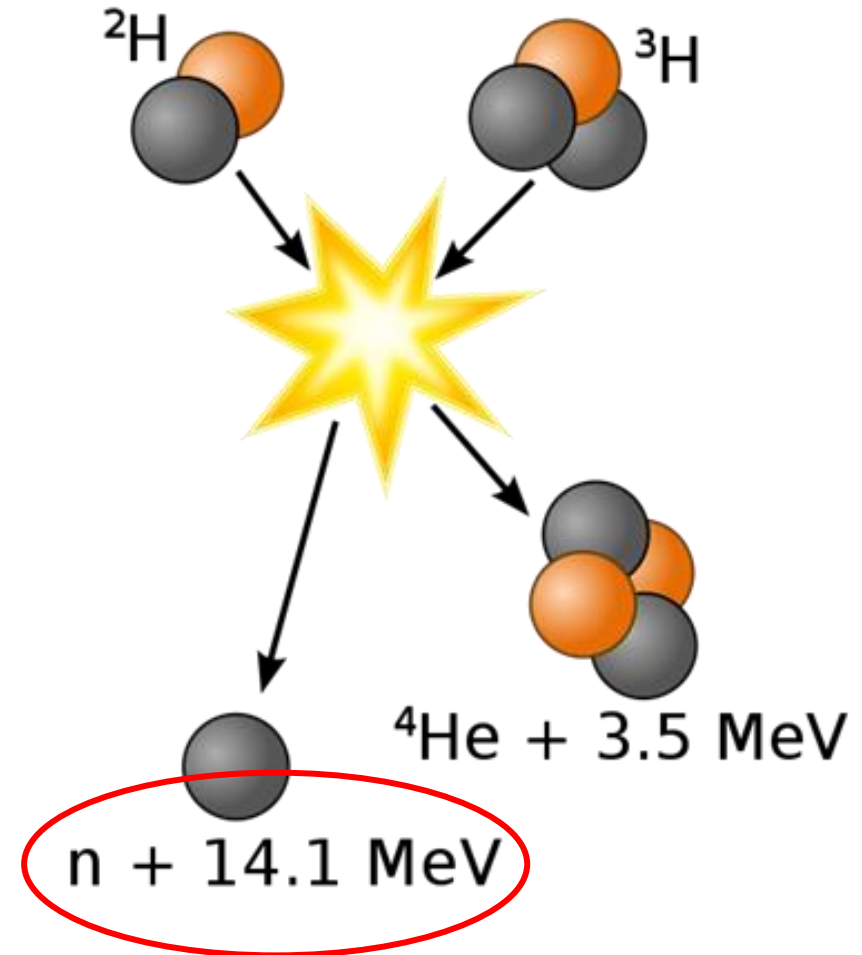
In Space:

Intense heat & pressure due to gravity
~15 million K (core), 5,500K (surface)



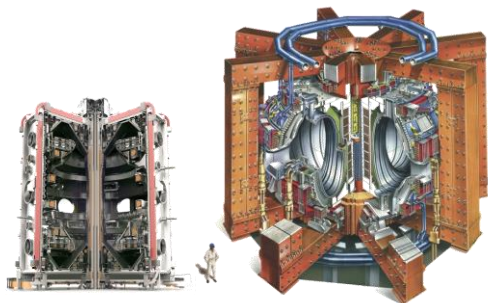
On Earth:

Need plasma temp. of ~150 million K
+ pressure (confinement)



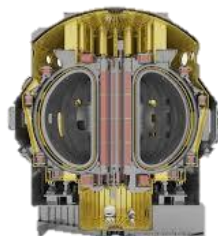
Fusion Reactors (Tokamaks)

A donut shaped magnetic bottle for confining and heating plasma to create fusion

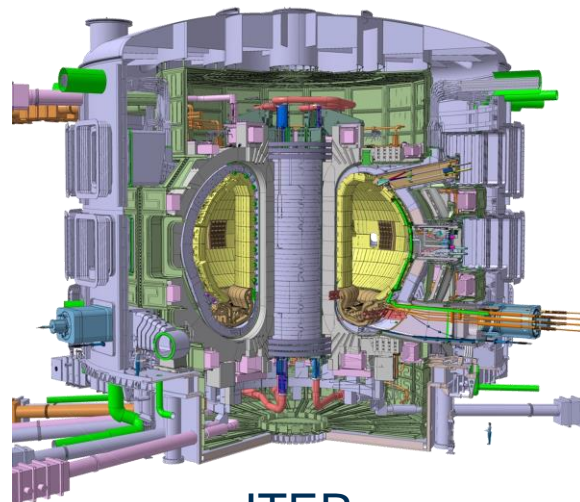


MAST-U

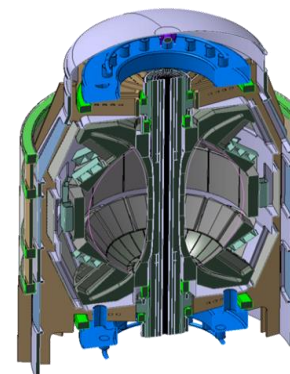
~~JET~~
Shutdown end 2023



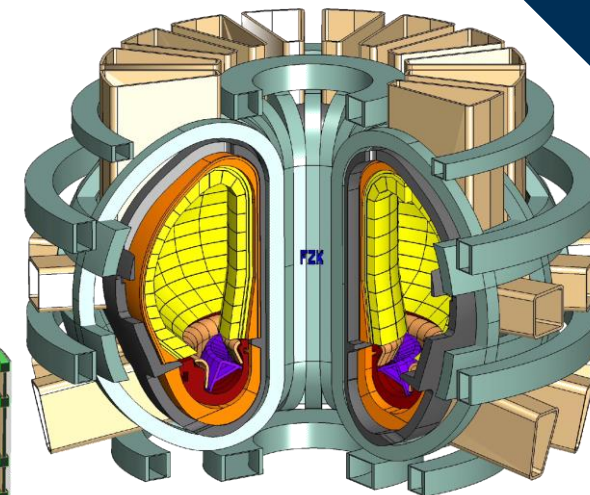
JT60-SA



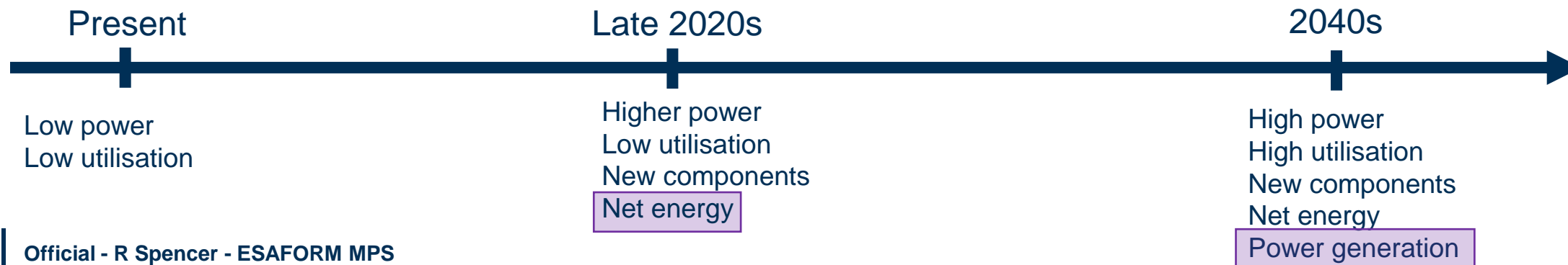
ITER



STEP



DEMO



Thermal Transients

Extract heat & produce tritium

Very Cold! (~ 4 K)

Ceramics & Composites

Field-Assisted Sintering



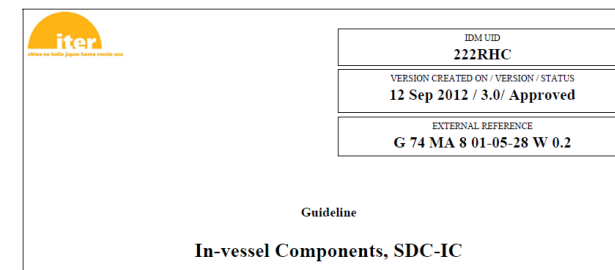
Design Data



Existing Design Codes
ASME BPVC
R5/6
RCC-MRx

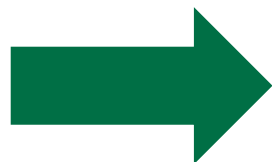


Fusion Design Codes
ITER SDC-IC
DEMO DC



Failure Modes
Plastic Collapse
Fatigue
Ratchetting
Creep
Local Failure
Fast Fracture
Creep-Fatigue

Needs data over

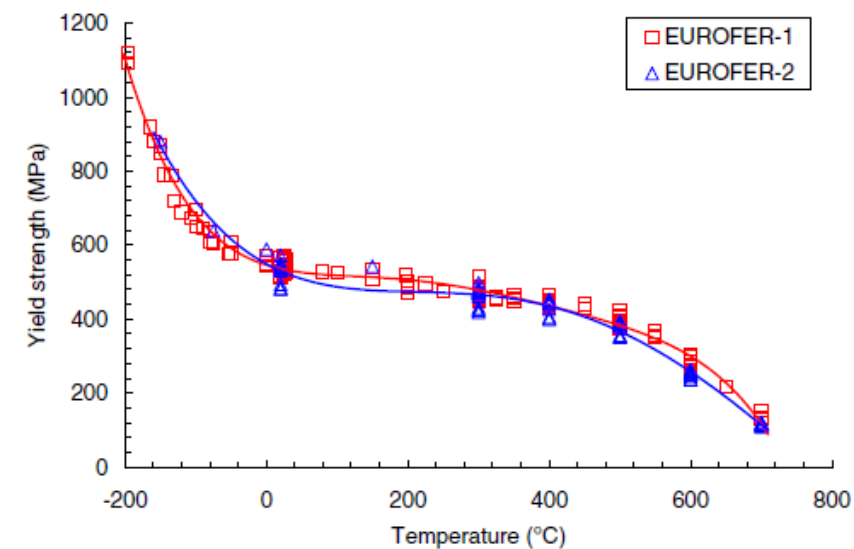


Temperature

Stress

Irradiation Damage

For Eurofer97 this
has taken >20 years



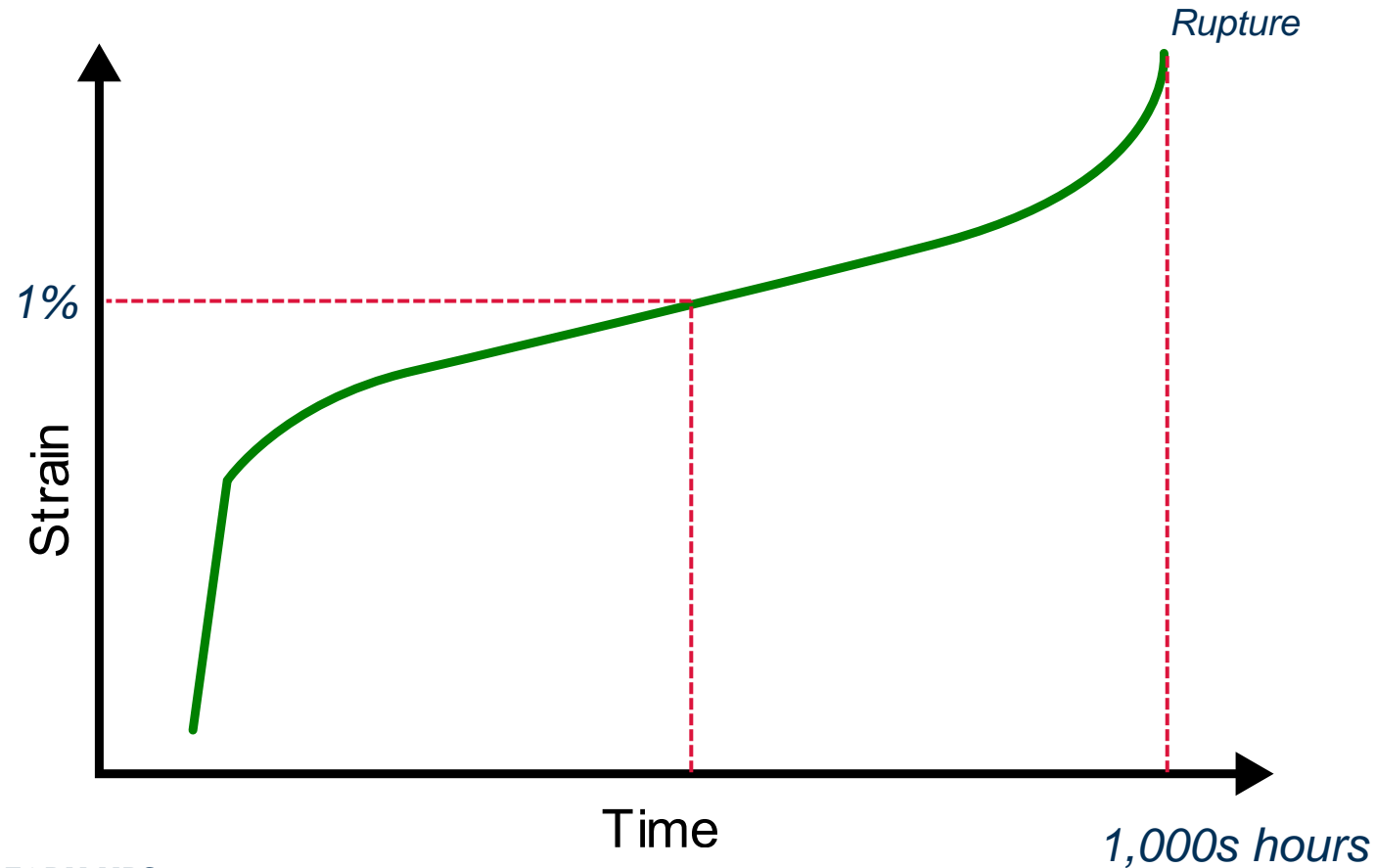
*Eurofer97 Yield Strength vs
temperature (unirradiated)
Lucon & Vandermeulen 2009*

Why focus on creep?

Creep is a key failure mode

Higher temperatures, higher efficiency, worse creep

Development of new radiation tolerant, creep-resistant steels

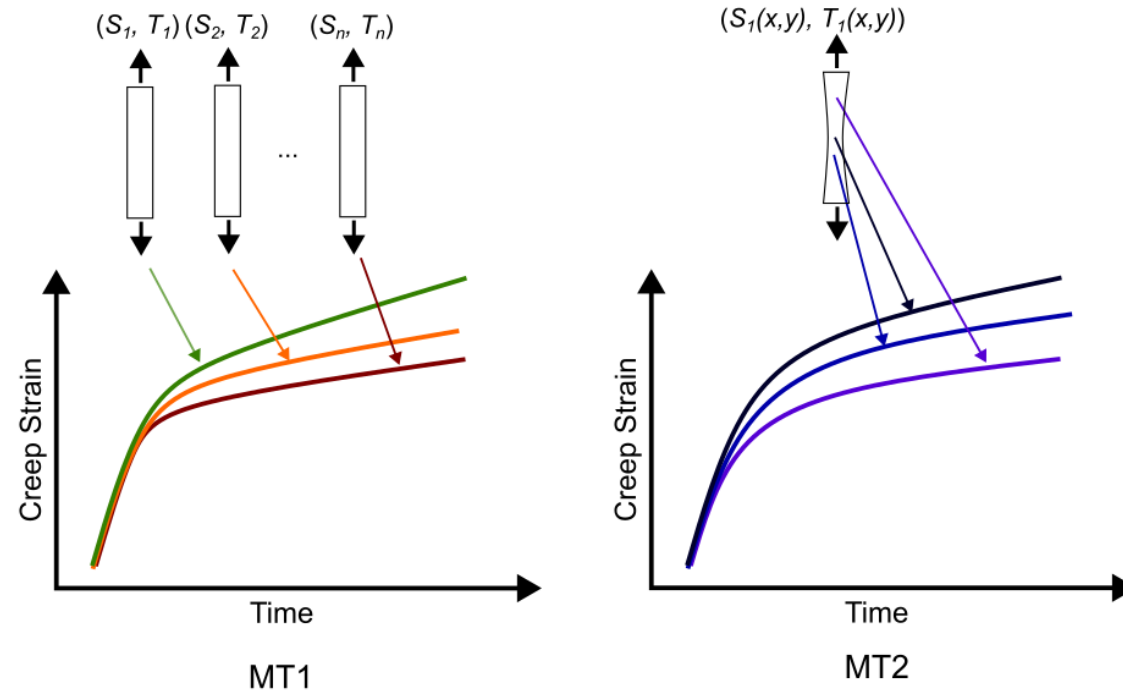


Why Materials Testing 2.0?

Material Testing 2.0 uses **complex tests**, **full-field measurements** and **inverse identification** to determine constitutive models.

Creep tests are long – 1,000's of hours, machines, time & resources to do them are expensive.

Can we use the Materials Testing 2.0 philosophy to get more data from a reduced number of tests?



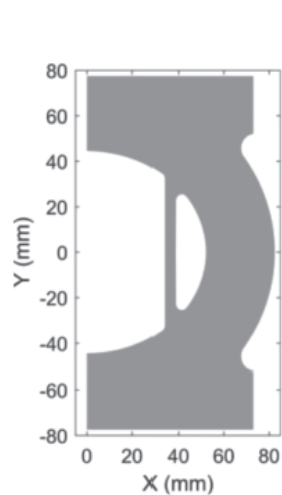
Does data from MT2 tests match that from MT1 tests?

Exploiting MT2: Test Design

In MT2 the test design space is now **infinite**, so how do you choose a suitable geometry? From [1]:

Intuition

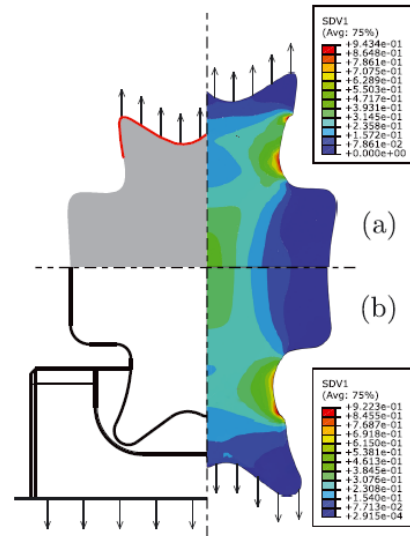
What do we think would work?



Viscoplastic Model Specimen [2]

Strain State

What gives the best spread of strain states?



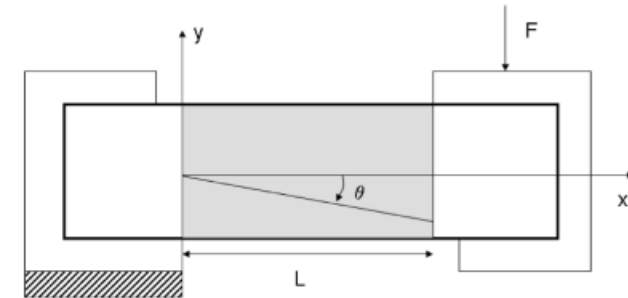
Anisotropic Model Specimen [3]

Identification Quality

What gives the least uncertainty on outputs?

Full Simulation

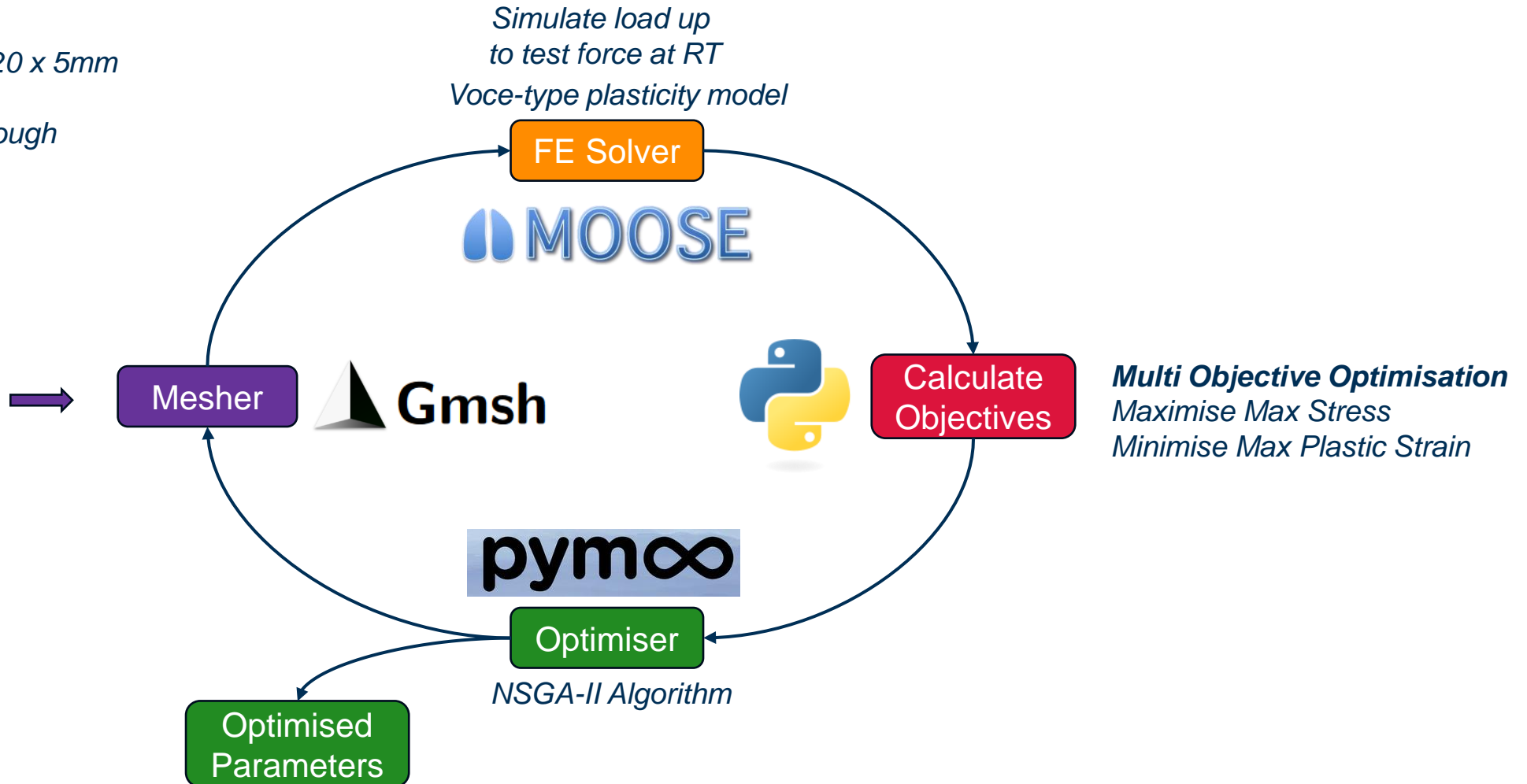
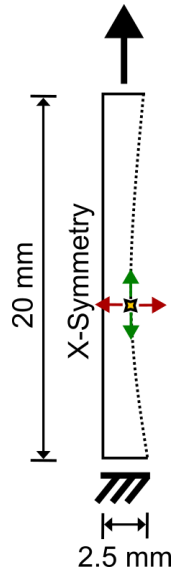
What is the effect of the measurement system?



Anisotropic Elasticity Test [4]

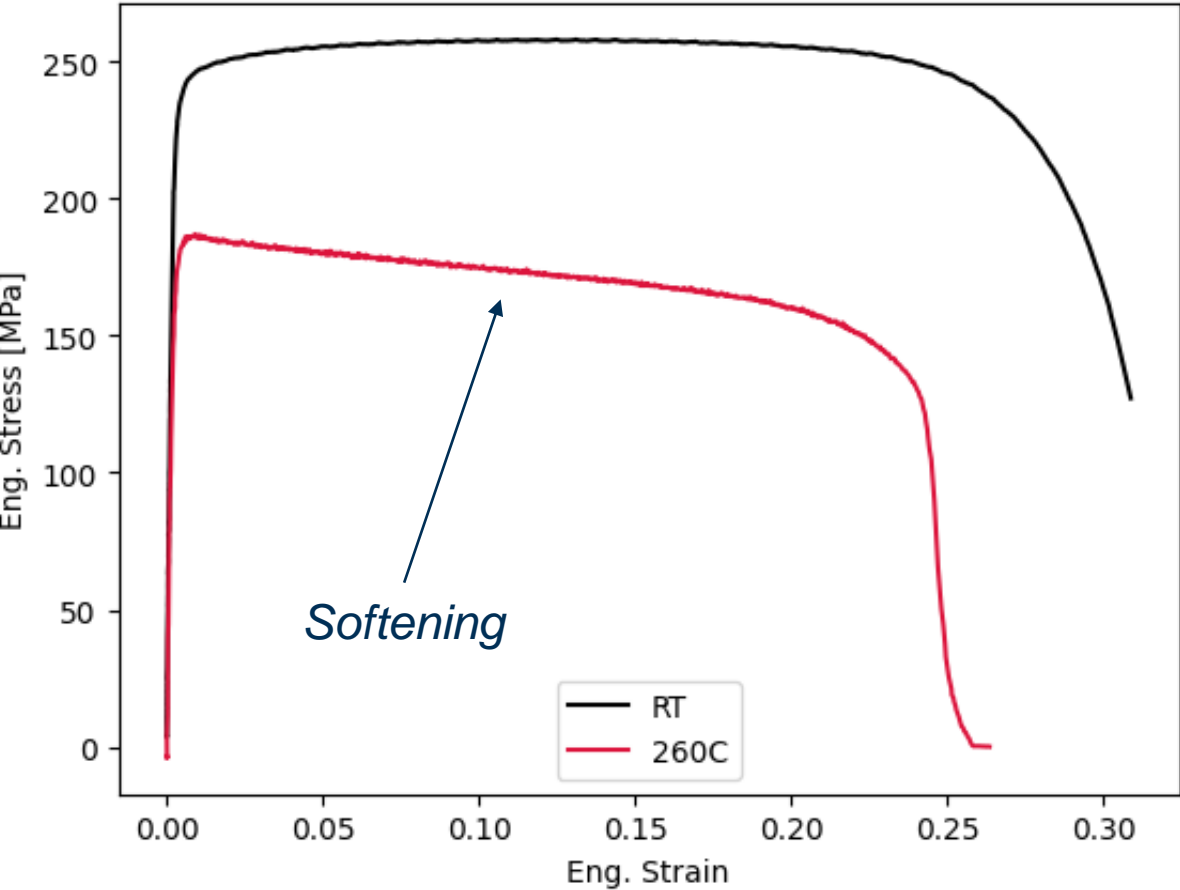
'Stress State' Optimisation Strategy

Geometry envelope - 20 x 5mm
Free moving point
Spline constructed through

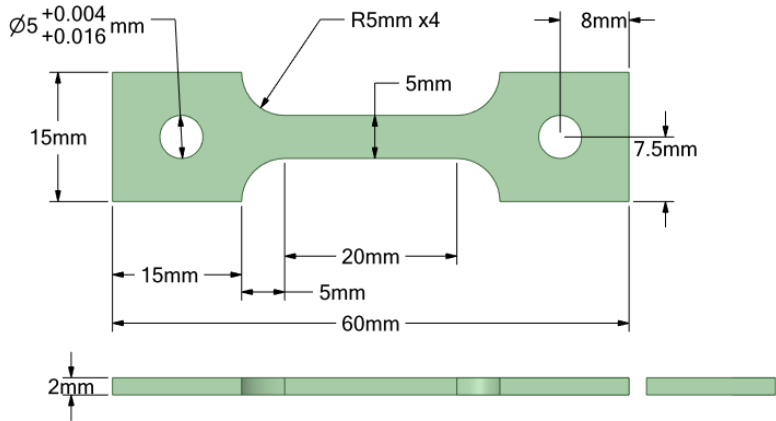


Material & Uniaxial Tensile Tests

OFHC Cu – Rolled Sheet, ½ Hard
 Proposed for interlayer between CuCrZr pipes and W armour



Temperature [C]	Elastic Modulus [GPa]	Yield Strength [MPa]	Tensile Strength [MPa]	'Uniform Elongation'
20	110	160	258	0.125
260	98	122	122	0.009



Based on ASTM-E8

MT2 Geometry

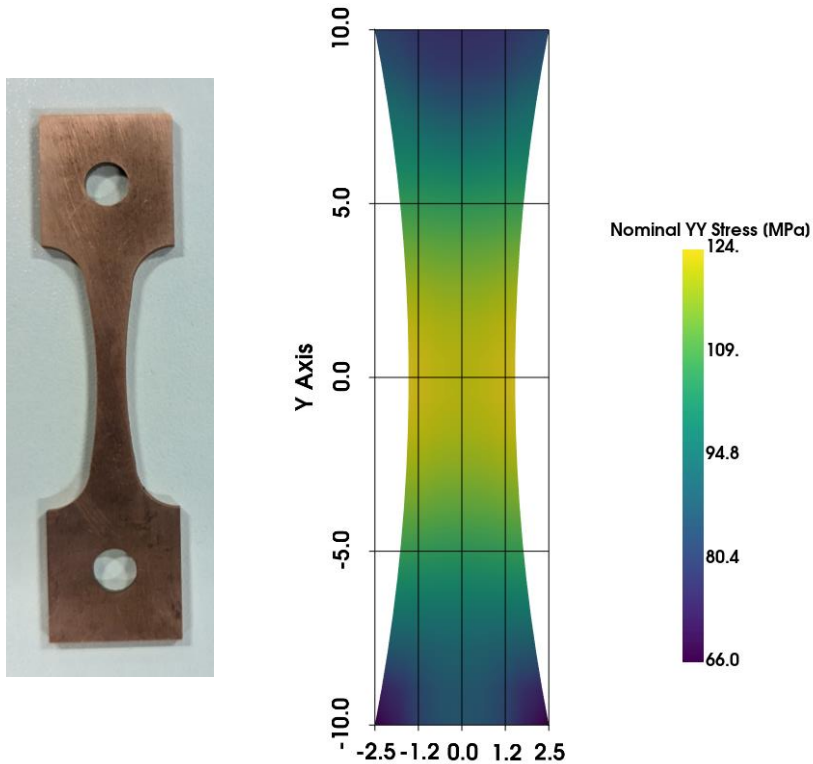
MT2 Specimen

MT2 tapered specimen geometry

Load - 750N

Approx Stress range 70-125MPa

Temp ~ 300°C



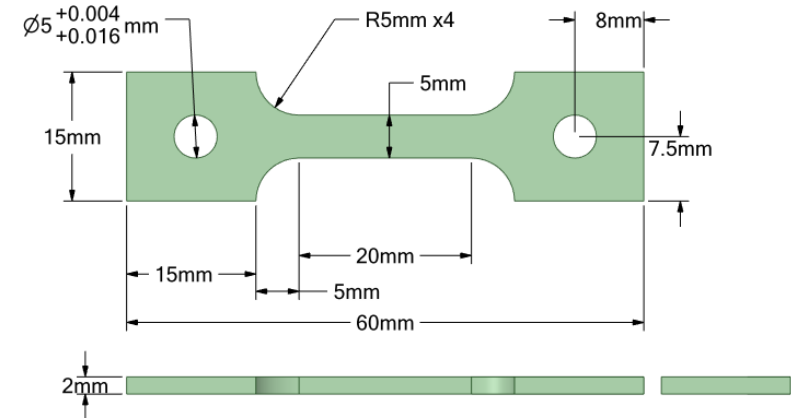
Uniaxial Specimens

ASTM E8 type specimen, ~20% subsize

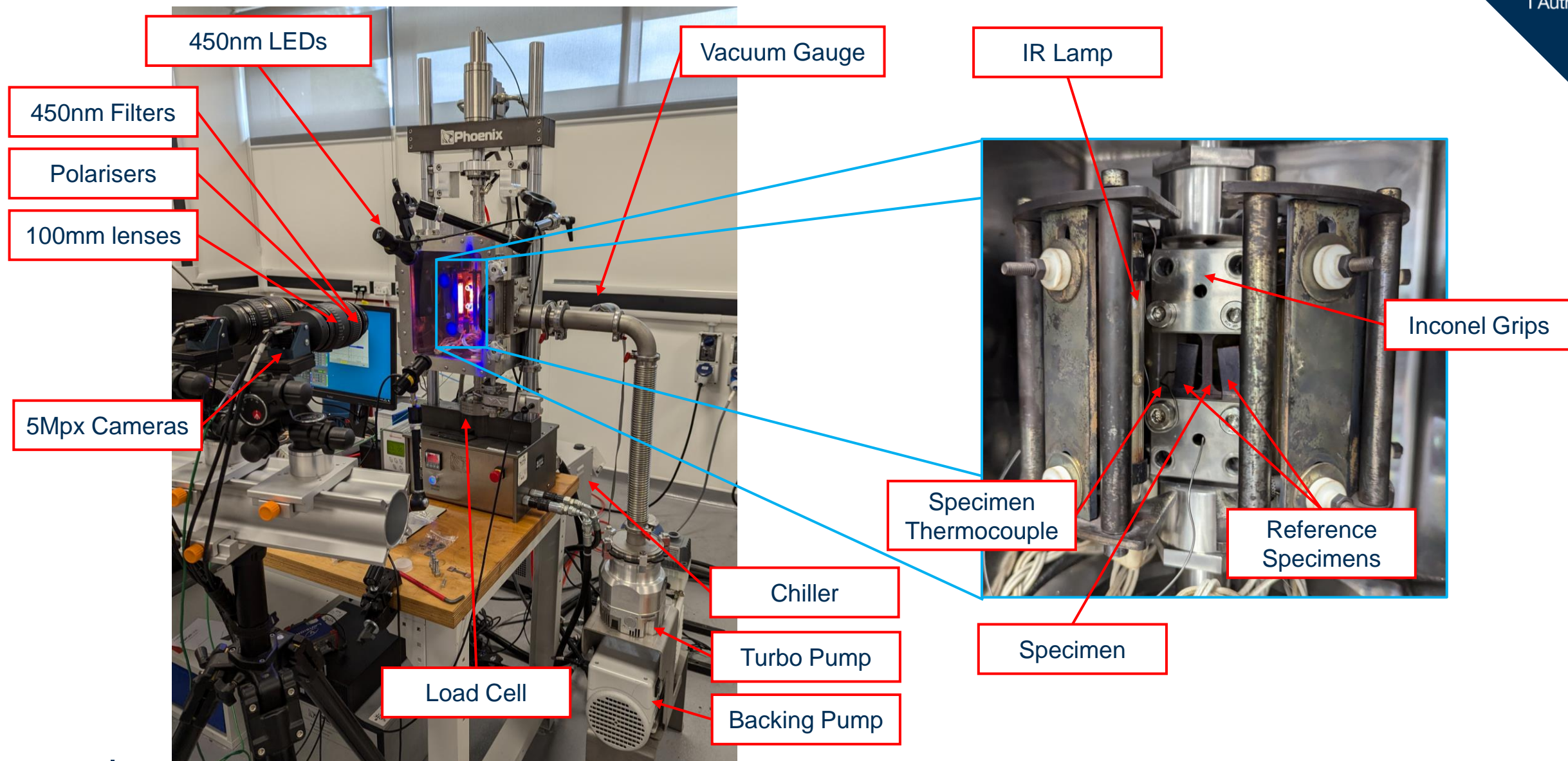
Loads – 800, 1000, 1200N

Stresses – 80, 100, 120MPa

Temp ~ 300°C

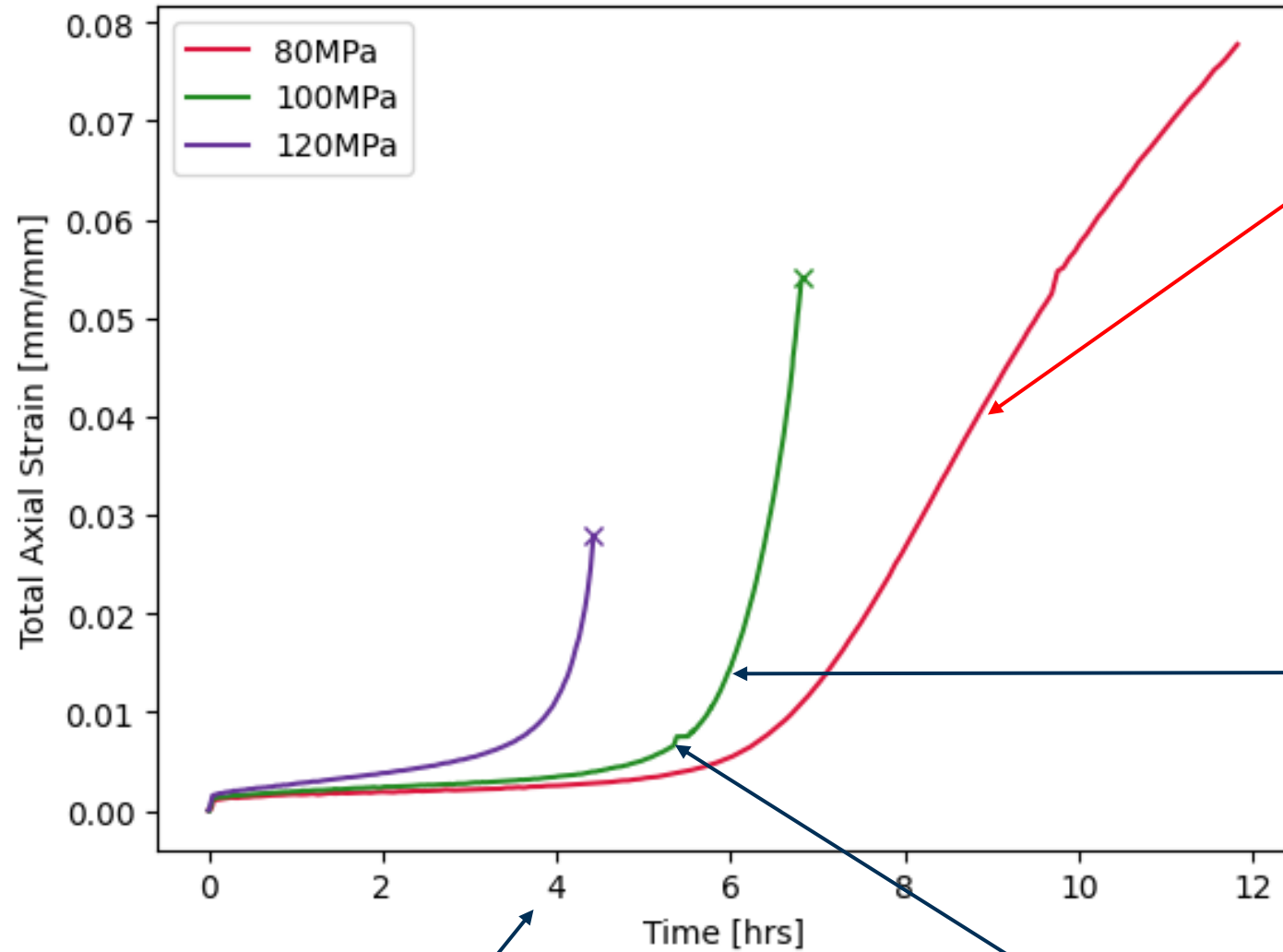


Experimental Setup



MT1 Creep Tests

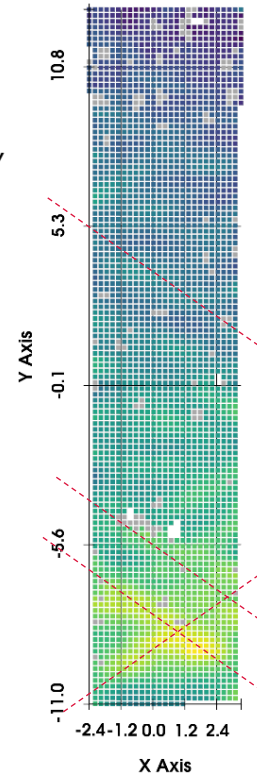
Creep of ½ Hard OFHC Cu at 265°C



Thermal drift?

100MPa, t=6hrs

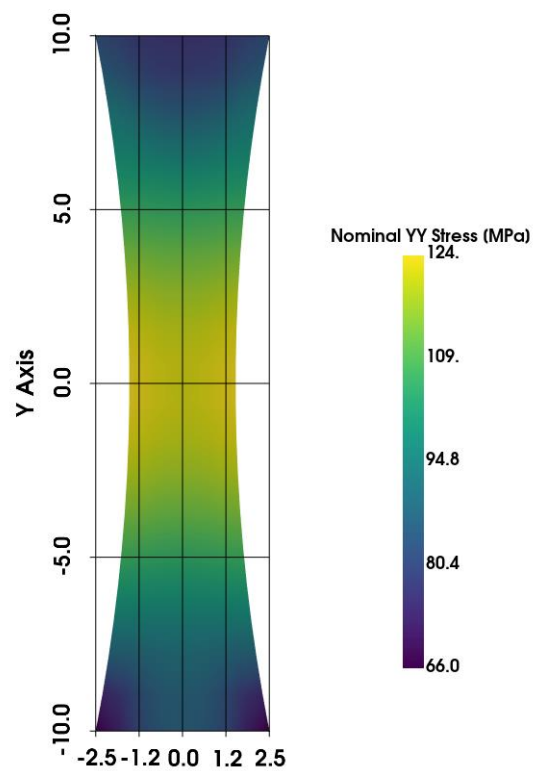
Total Strain YY
0.0292
0.0225
0.0159
0.0093
0.0027



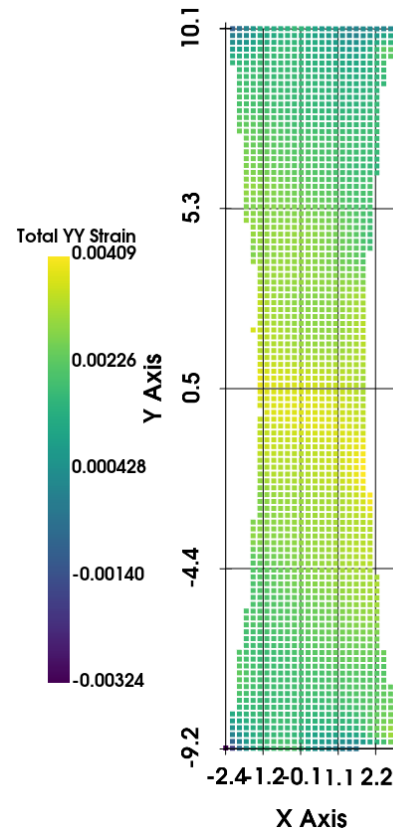
High stress, short test time, iterate faster

DIC Control PC fell over

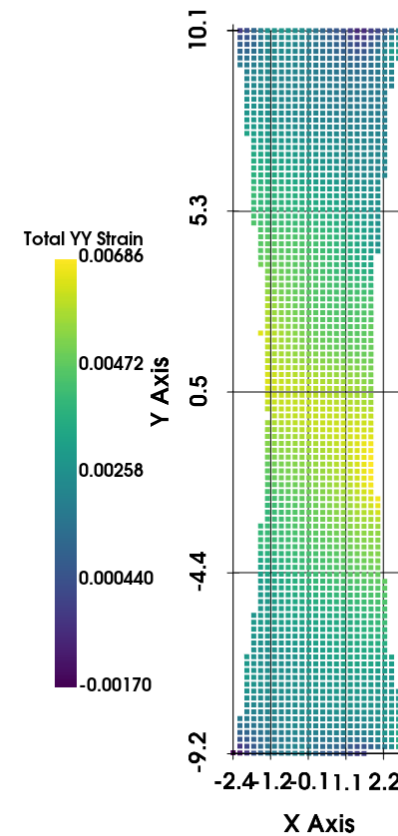
MT2 Specimen Results



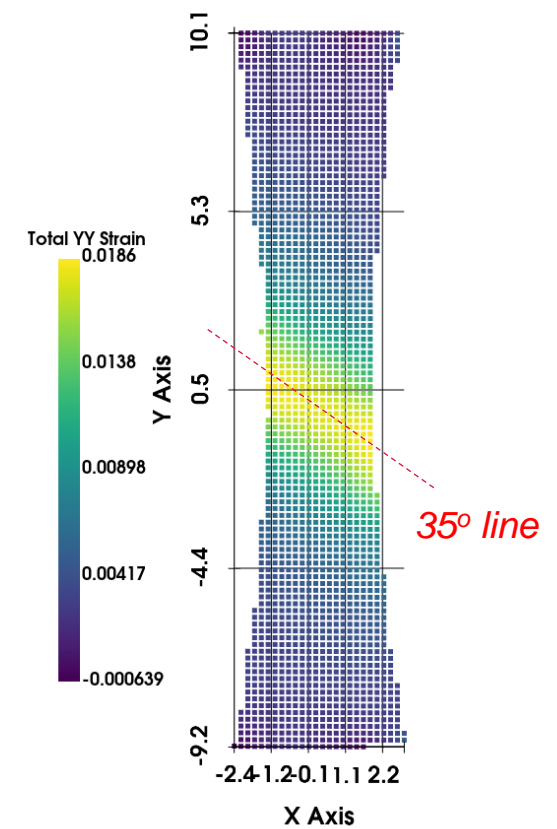
*Nominal stress after
elastic loading*



t=1.5hrs

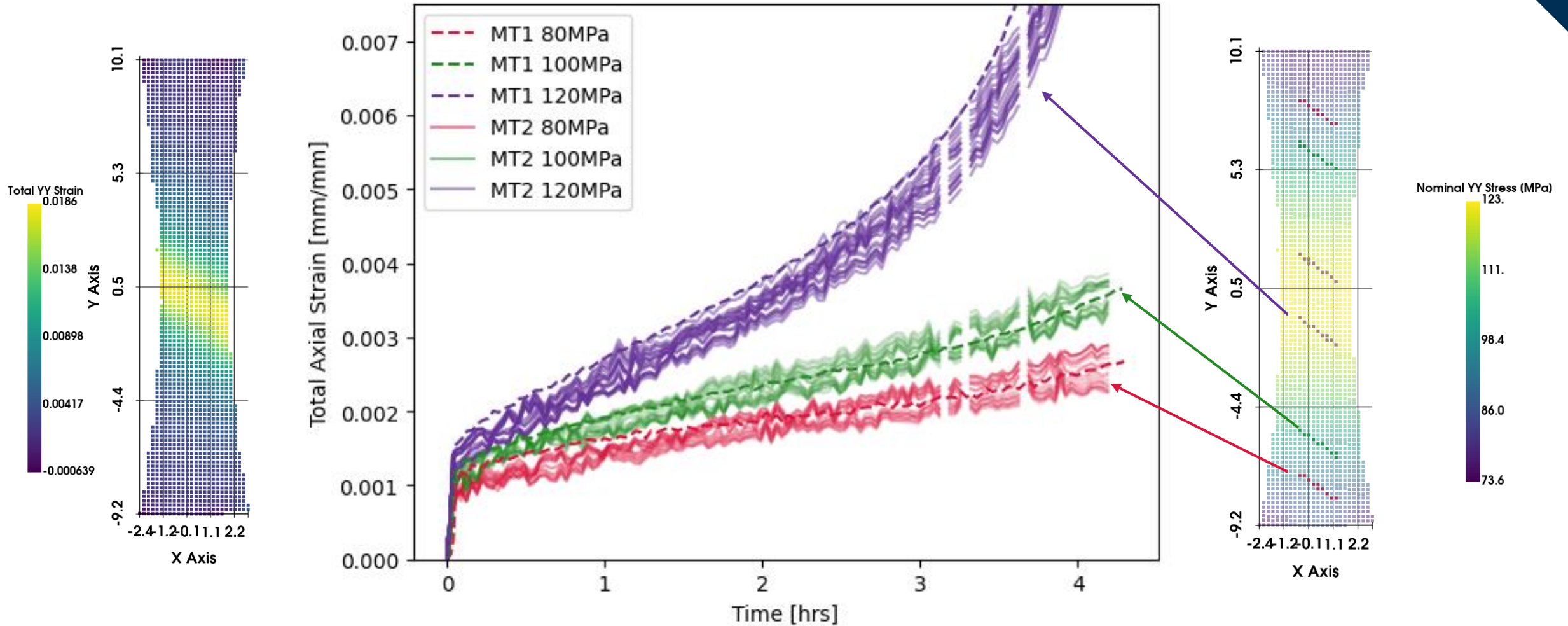


t=3hrs



t=4.18hrs

MT1-2 Comparison



Modelling

- Isotropic model (neglect anisotropy)
- Unified Viscoplasticity with Damage
- Damage degrades elastic stiffness
- Implemented in MOOSE Open-Source FE Solver

Strain Partitioning

$$\varepsilon = \varepsilon_{el} + \varepsilon_{vp}$$

Rate Sensitivity / Viscoplasticity

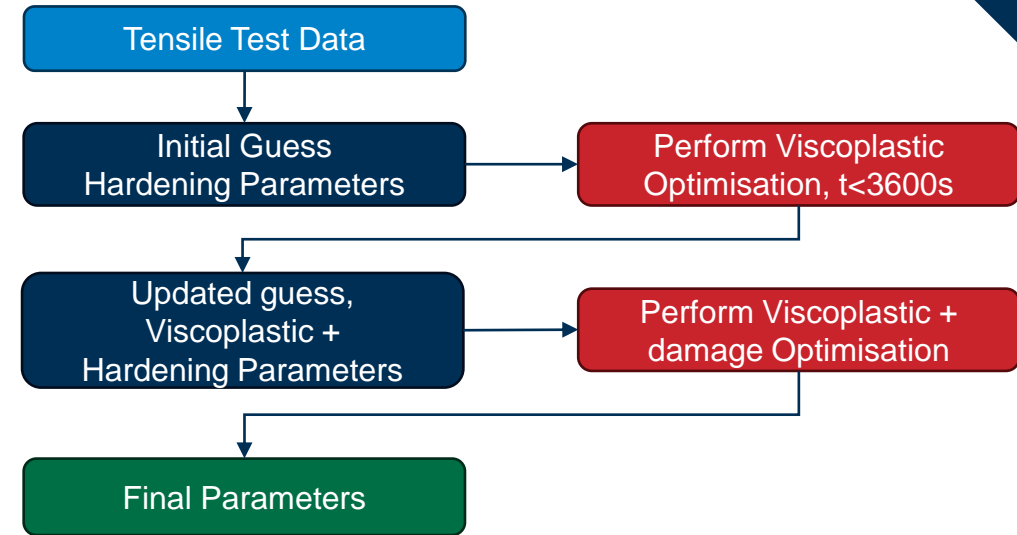
$$\dot{\varepsilon}_{vp} = \alpha \sinh \beta \left(\frac{\sigma}{1 - \omega} - R - \sigma_y \right)$$

Voce Strain Hardening

$$R = \sigma_s (1 - e^{b\varepsilon_{vp}}) + h\varepsilon_{vp}$$

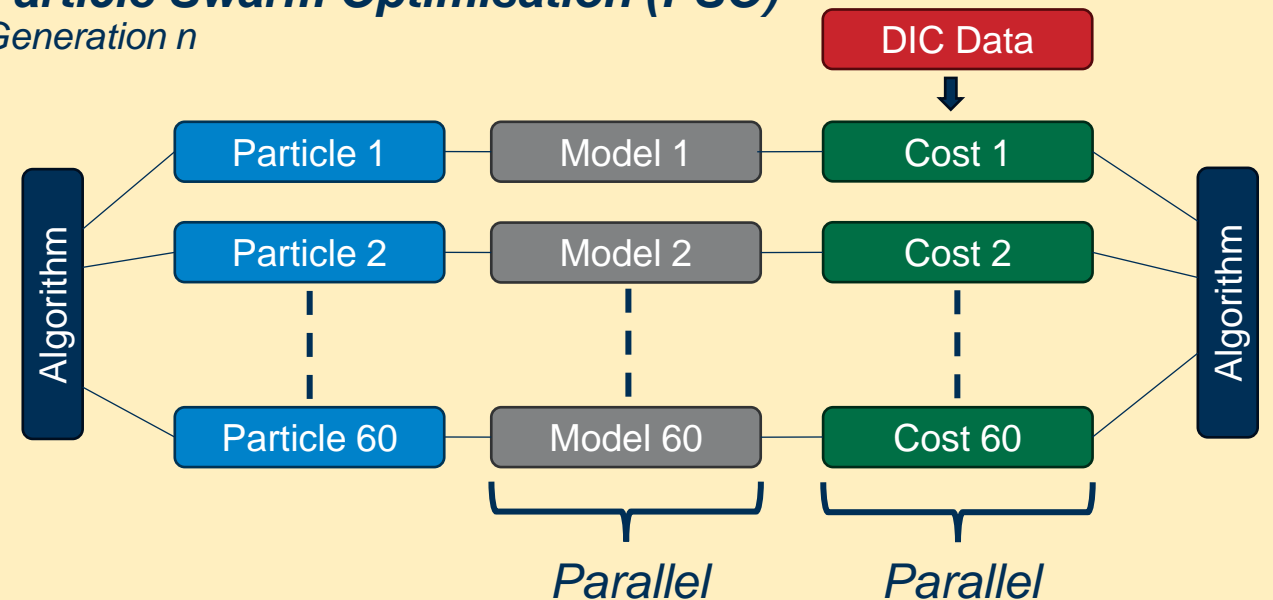
Leckie & Hayhurst Damage

$$\omega = 1 - \left[1 - \frac{t}{\left(\frac{\sigma}{A} \right)^{-\zeta} (1 + \phi)^{-1}} \right]^{\frac{1}{1+\phi}}$$



Particle Swarm Optimisation (PSO)

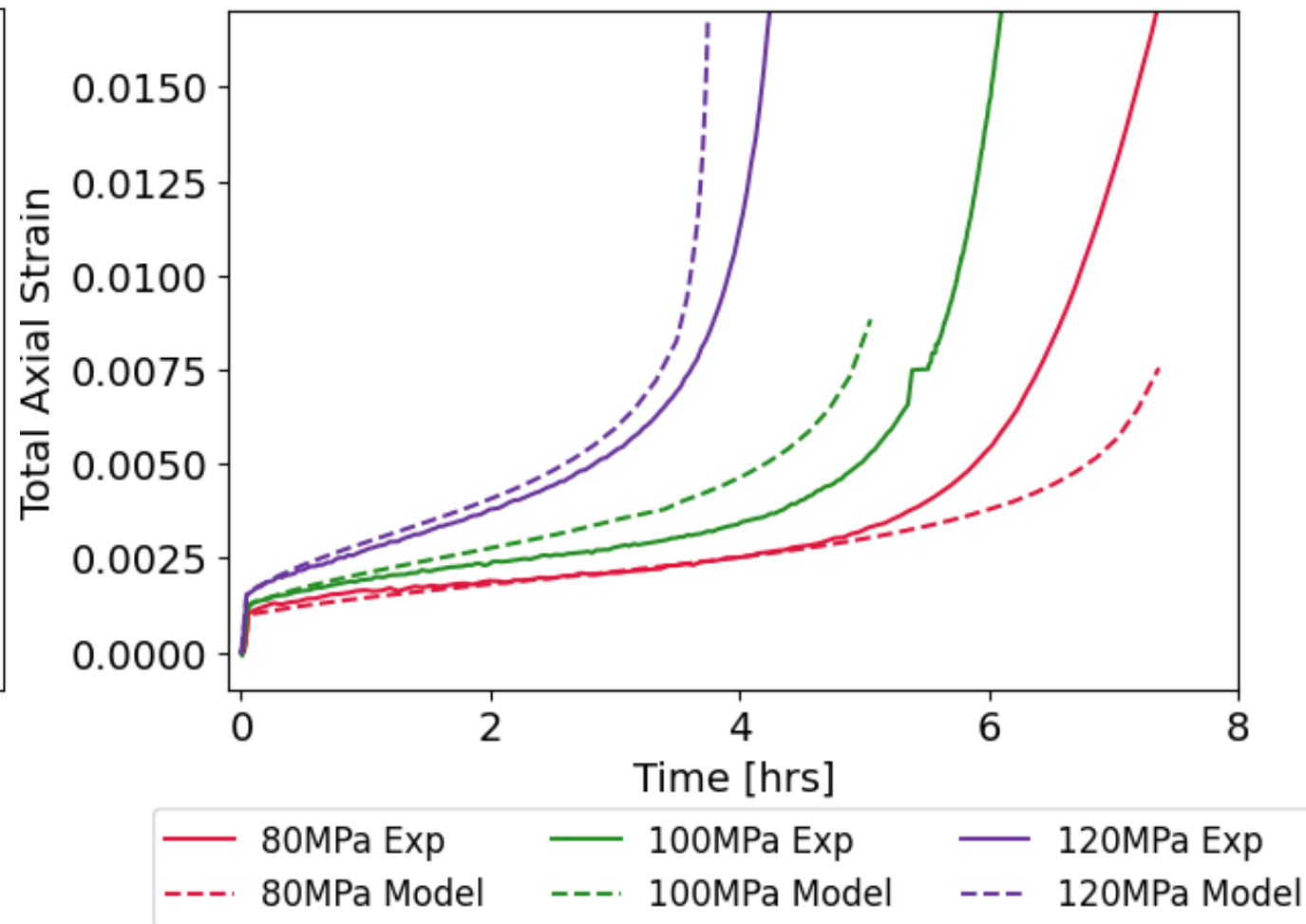
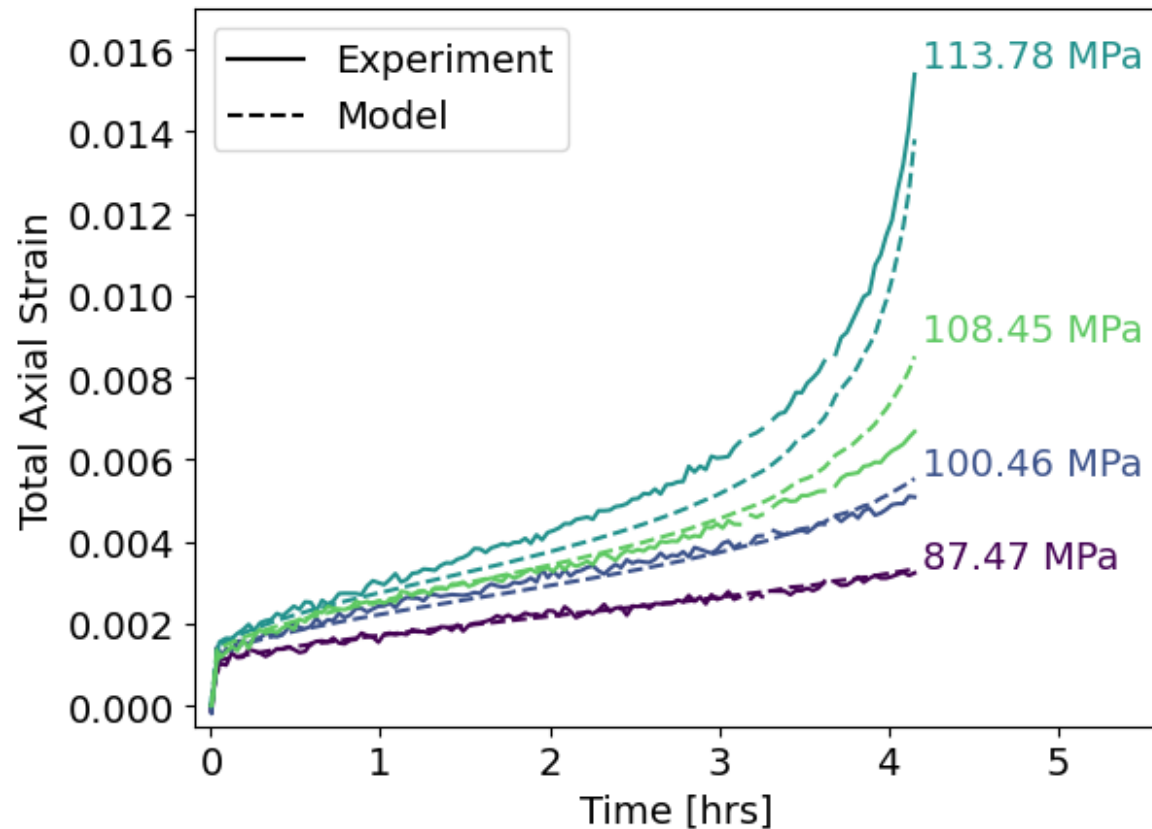
Generation n



Modelling Results

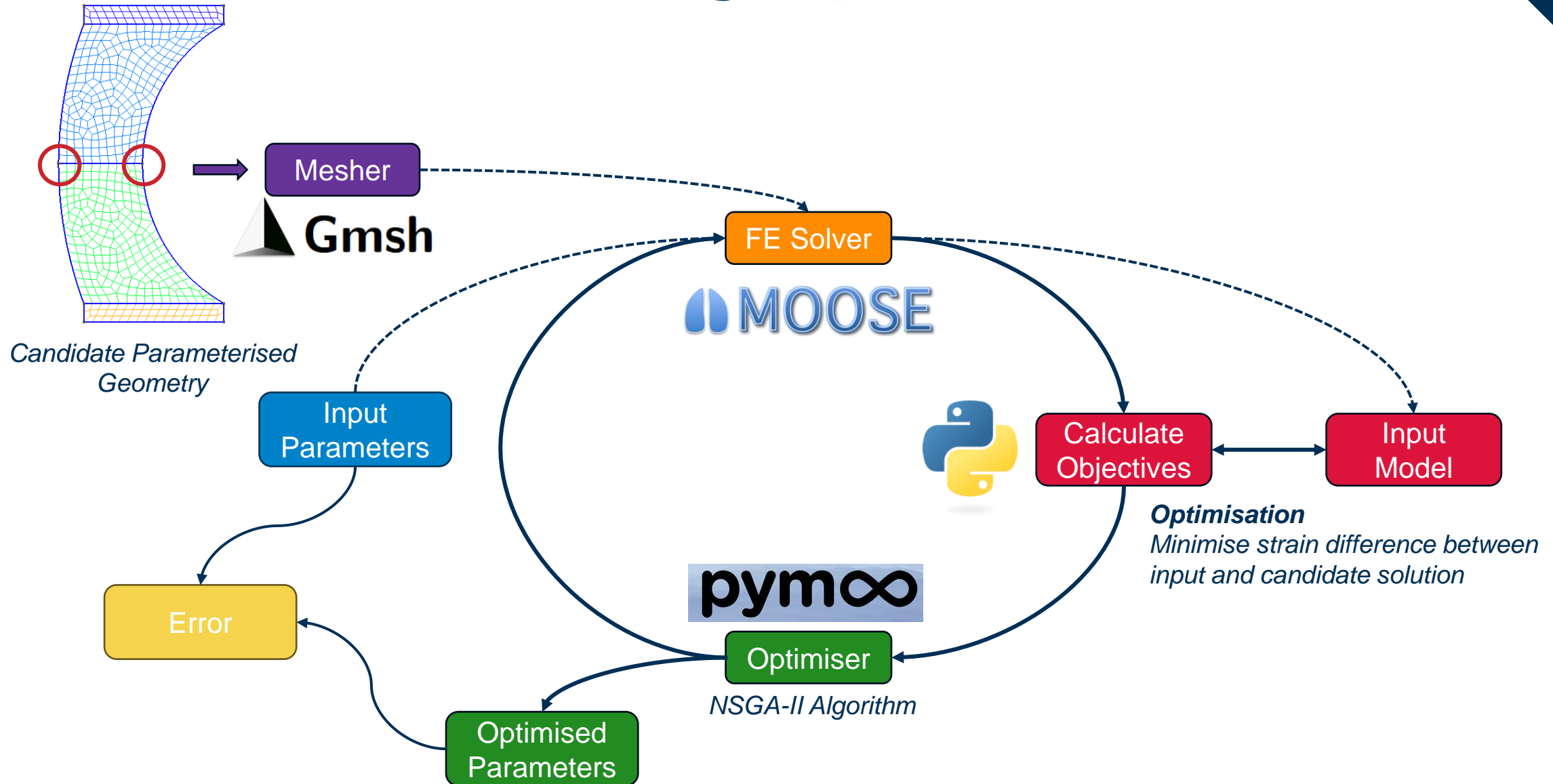
MT2

Uniaxial

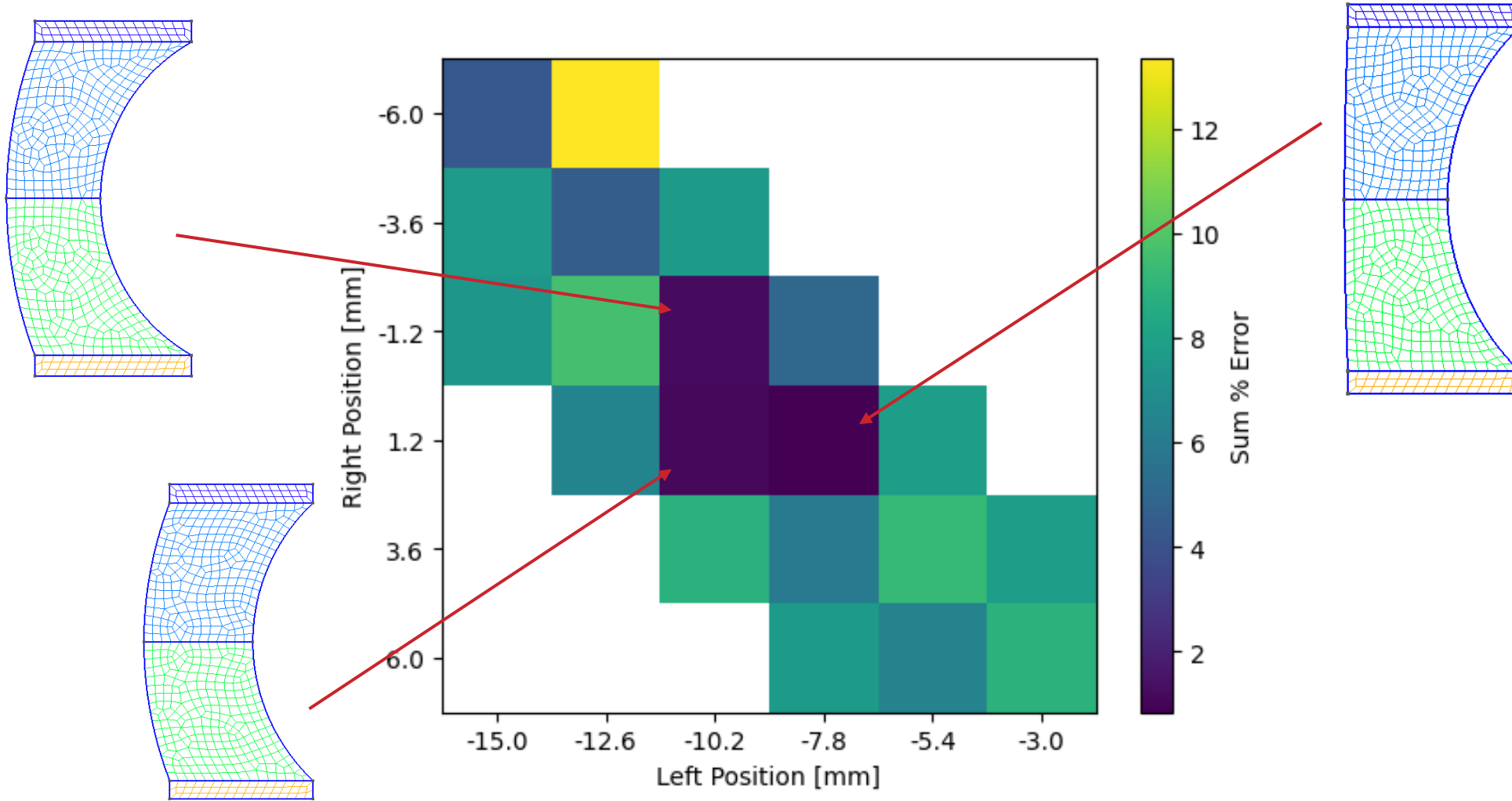


- Agreement is promising, given lack of anisotropy
- Experimental temperature inaccuracy

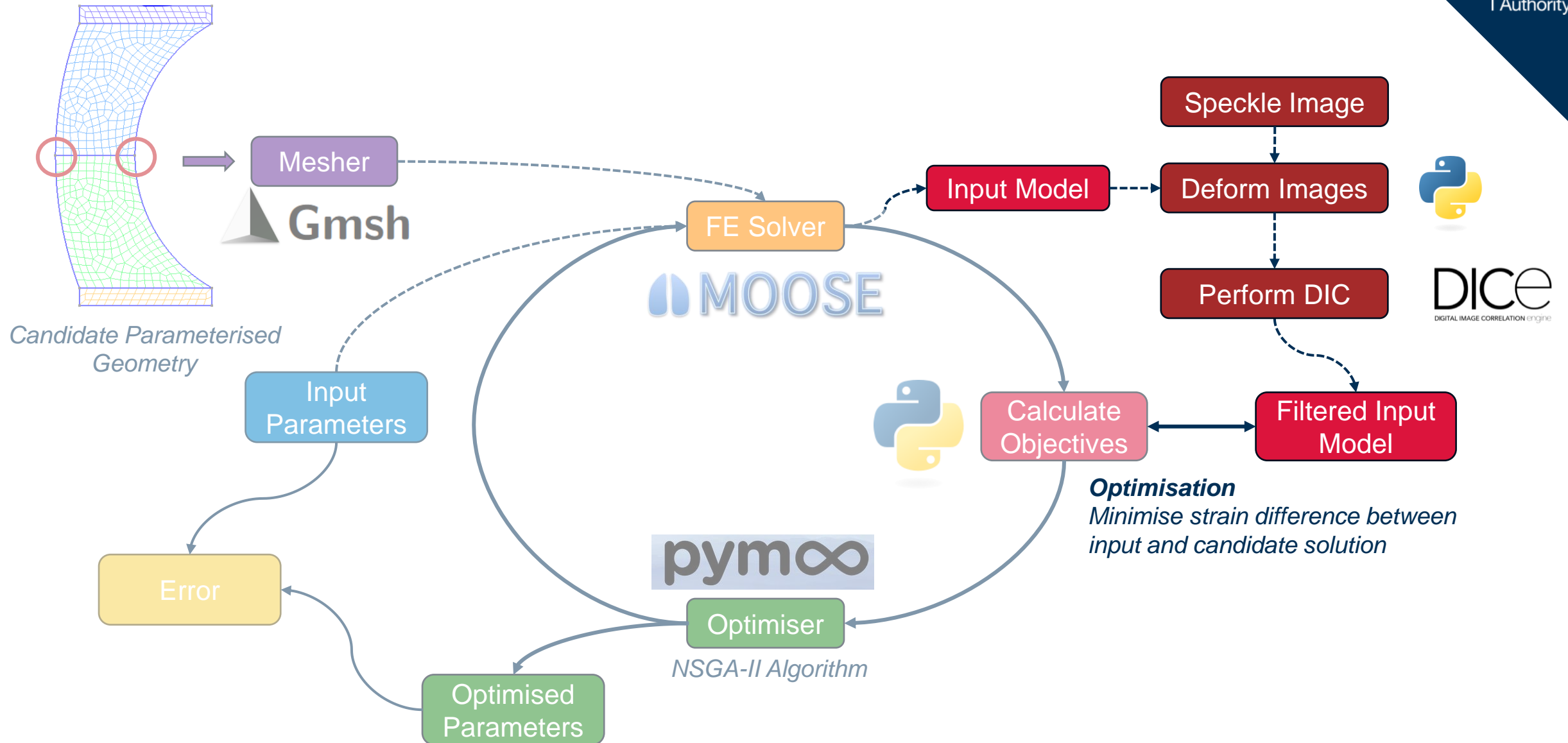
Identification Quality Optimisation



Identification Quality Grid Search



Towards Full Simulation



Conclusions & Outlook

Tested novel geometry & uniaxial specimens at high temperature with similar creep stresses

Demonstrated multiple creep curves from one specimen at high temperature

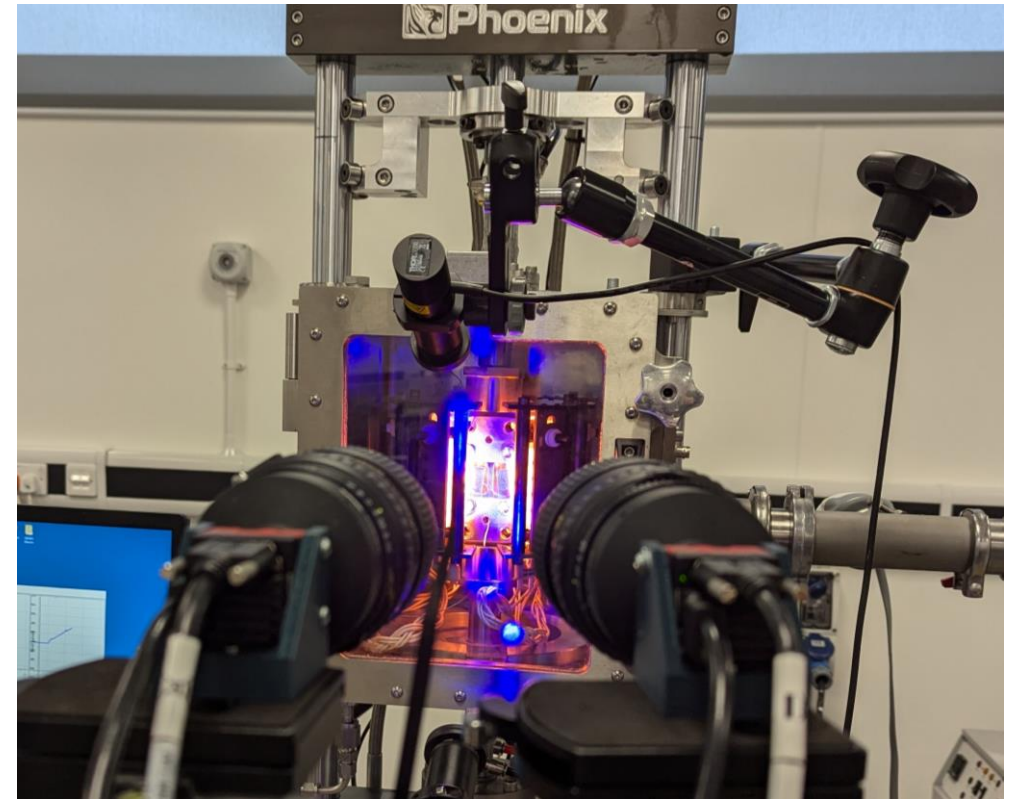
Fitted isotropic model to data and demonstrated reasonable agreement with experiment

Going forward:

- Improved design optimisation

- Targeting Gr91 steel 600-700°C

- Data from uniaxial and MT2 geometries



Thank you

Any questions?

Contact: rory.spencer@ukaea.uk